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Investigation on microwave dielectric properties of new low-loss  $\text{CoZrTa}_2\text{O}_8$  ceramicsYing Wang<sup>†</sup>, Shao-Bo Zhang, Tian-Liang Tang, Wang-Suo Xia, Li-Wei Shi<sup>†</sup>

School of Physical science and technology, China University of Mining and Technology, Xuzhou

221116, China

**Abstract**

New ceramics with  $\text{CoZrTa}_2\text{O}_8$  formula were prepared by conventional solid-state reaction method. The phase composition of all sintered samples was confirmed as a single phase with monoclinic wolframite structure belonged to the  $P2_1/c$  space group. The grain growth was analyzed through the investigation of relative density and grain size. The dielectric constant ( $\epsilon_r$ ) was discussed with relative density, polarizability and porosity. The quality factors ( $Q \times f$ ) were analyzed by considering grain growth and atomic packing fraction. The temperature coefficient of resonant frequency ( $\tau_f$ ) had no significant change under different sintering temperature. The typical microwave dielectric properties of  $\text{CoZrTa}_2\text{O}_8$  ceramics were  $\epsilon_r=23.54$ ,  $Q \times f=20100\text{GHz}$ , and  $\tau_f=-8.72\text{ppm}/^\circ\text{C}$ , sintered at  $1250^\circ\text{C}$  for 4h. Considering the low dielectric loss and near-zero  $\tau_f$  value,  $\text{CoZrTa}_2\text{O}_8$  ceramics should be good candidates for microwave device application.

**Keywords:** Ceramics; Microwave dielectric properties; Sintering; Structural feature

**1. Introduction**

With the ongoing revolution in wireless communication, microwave dielectric ceramics, which play an important role in microwave devices, have attracted increasing research interests [1-4]. Under this background, many researchers have been committed to research and develop new microwave dielectric

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<sup>†</sup> Corresponding author. Tel./fax: +86 18852140230

E-mail address: [wangy@cumt.edu.cn](mailto:wangy@cumt.edu.cn) (Y. Wang), [liwei5450@cumt.edu.cn](mailto:liwei5450@cumt.edu.cn) (L. W. Shi)

ceramics.

Zn-AO<sub>2</sub>-Nb<sub>2</sub>O<sub>5</sub> (A= Ti, Zr, Sn) system had been studied firstly by A. Baumgarte and R. Blachnik in 1994 [5]. In that paper, the crystalline structure of ZnTiNb<sub>2</sub>O<sub>8</sub> had been reported with orthorhombic, space group *PBCN*. Then, the phase composition and microwave dielectric properties of Zn<sub>3</sub>Nb<sub>2</sub>O<sub>8</sub>-TiO<sub>2</sub> and ZnNb<sub>2</sub>O<sub>6</sub>-TiO<sub>2</sub> had been studied by D. W. Kim *et al* [6, 7]. In 2011, ZnTiNb<sub>2</sub>O<sub>8</sub> ceramics, as a new microwave dielectric material, had drawn a great of interests owing to the excellent microwave dielectric properties:  $\epsilon_r \sim 34$ ,  $Q \times f \sim 50000 \text{ GHz}$ ,  $\tau_f \sim -45 \text{ ppm/}^\circ\text{C}$  [8]. Thereafter, a series of novel microwave dielectric ceramics, which had similar formulate to ZnTiNb<sub>2</sub>O<sub>8</sub>, had been reported for their possible use in microwave application [9-12]. Among the researches, it was important to note that those new ceramics were almost obtained by ionic substitution on Zn-site, Ti-site or Nb-site of ZnTiNb<sub>2</sub>O<sub>8</sub> ceramics. However, what was noteworthy was that there were less investigations on A<sup>2+</sup>ZrTa<sub>2</sub>O<sub>8</sub> system, which obtained by employing Zr and Ta ions as substitution ions. Especially, there were no report on A<sup>2+</sup>ZrTa<sub>2</sub>O<sub>8</sub> ceramics with A=Mn, Co, Ni. To develop a new-type microwave dielectric ceramic with excellent properties, and considering that Co ion was a good substitution in the type of bivalent ions, high-Q microwave dielectric ceramics CoZrTa<sub>2</sub>O<sub>8</sub>, which were made by utilizing Co to occupy A-site of A<sup>2+</sup>ZrTa<sub>2</sub>O<sub>8</sub> ceramics, had been prepared and investigated.

In this work, CoZrTa<sub>2</sub>O<sub>8</sub> ceramics, which exhibited a pure phase with monoclinic wolframite structure, were successfully prepared for the first time up to now. Moreover, microwave dielectric properties of CoZrTa<sub>2</sub>O<sub>8</sub> ceramics had been reported, with the influence of intrinsic factors and extrinsic factors on microwave dielectric properties being taken into overall consideration.

## 2. Experimental procedure

CoZrTa<sub>2</sub>O<sub>8</sub> ceramics were synthesized using conventional mixed-oxide methods with high-purity

oxide powders including CoO (99%), ZrO<sub>2</sub> (99%) and Ta<sub>2</sub>O<sub>5</sub> (99.9%). These powders were milled under distilled water for 12h with zirconia ball. After milling, the slurries were dried using a dry oven and sieved with a 40 mesh screen. The powders were calcined at 1000°C for 4h in an alumina crucible. Then the calcined powders were re-milled for 24h in distilled water. Next, the slurries were dried and sieved again with an 80 mesh screen. The final powders were pressed into pellets with 10 mm in diameter and 5 mm in thickness. Ultimately, these pellets were sintered between 1175°C and 1275 °C for 4h in air.

Crystalline phases of CoZrTa<sub>2</sub>O<sub>8</sub> ceramics were analyzed by powder X-ray diffraction analysis using a Rigaku diffractometer (Model D/Max-B, Rigaku Co., Japan) with Ni filtered Cu K $\alpha$  radiation ( $\lambda=0.1542\text{nm}$ ) in the  $2\theta$  range of 10-90°. The ceramic surface was observed and analyzed using a scanning electron microscopy (SEM, FEI Quanta 250, USA). Densities of the sintered specimens were measured using the liquid Archimedes method (Mettler Toledo XS64) with distilled water. Moreover, the microwave dielectric properties were measured via Hakki-Coleman's dielectric resonator method using a network analyzer (N5234A, Agilent Co., USA) in the frequency range of 7-10 GHz [13, 14]. The  $\tau_f$  values were calculated with temperature variation of the TE<sub>016</sub> mode (temperature range from 25°C to 85°C). The  $\tau_f$  (ppm/°C) was defined as the following formula:

$$\tau_f = \frac{f_2 - f_1}{f_1(T_2 - T_1)} \quad (4)$$

where  $f_1$  was the resonant frequency at  $T_1$ , and  $f_2$  was the resonant frequency at  $T_2$ .

### 3. Result and discussion

Fig. 1 showed the SEM images of CoZrTa<sub>2</sub>O<sub>8</sub> ceramics sintered at different temperatures. There was no pore observed in all images, suggesting that the well-developed microstructure was obtained. The grain size of CoZrTa<sub>2</sub>O<sub>8</sub> ceramics increased from 0.45 $\mu\text{m}$  to 1.76 $\mu\text{m}$  with the increase of sintering

temperature, and uniform grain morphology was produced at 1250°C with the grain size of 0.94 $\mu\text{m}$ . As shown in Fig. 1(d), the abnormal grain growth occurred for  $\text{CoZrTa}_2\text{O}_8$  ceramics sintered at 1275°C, implying that excessive sintering temperature was unfavorable. As a result, the grain growth of  $\text{CoZrTa}_2\text{O}_8$  ceramics was good for their microwave dielectric properties when the sintering temperature ranged from 1175°C to 1250°C. When further increasing the sintering temperature, abnormal grain growth appeared, which had no benefit to microwave dielectric properties.

Fig. 2 showed the XRD patterns for the  $\text{CoZrTa}_2\text{O}_8$  ceramics sintered at 1175-1275°C for 4h. The diffraction patterns of  $\text{CoZrTa}_2\text{O}_8$  ceramics were indexed by wolframite structure type belonged to the space group  $P2_1/c$  ( $C_{2h}^4$ ), and no secondary phase was observed in all samples during the sintering process. In details, the peak position shifted to a lower angle when the sintering temperature increased, suggesting that the lattice parameters were affected by sintering temperature. Accordingly, the unit-cell volume of  $\text{CoZrTa}_2\text{O}_8$  ceramics was carried out by XRD data to investigate the effect of different sintering temperature. As shown in the illustrations of Fig. 2, the unit-cell volume of  $\text{CoZrTa}_2\text{O}_8$  ceramics increased obviously when the sintering temperature increased, which would affect the  $Q \times f$  values of  $\text{CoZrTa}_2\text{O}_8$  ceramics.

Fig. 3 illustrated the dielectric constant and relative density of  $\text{CoZrTa}_2\text{O}_8$  ceramics sintered at 1175-1275°C for 4h. Considering the sintering temperature effects, the dielectric constant of  $\text{CoZrTa}_2\text{O}_8$  ceramics was dependent on the relative density and phase composition. Refer to the analysis of XRD results, the relative density here was owed to pure crystalline phase. As shown in Fig. 3, the dielectric constant of  $\text{CoZrTa}_2\text{O}_8$  ceramics increased from 22.57 to 23.54 and then tended to be gentle, which was consistent with the variation of relative densities. In addition, it was known that the dielectric constant in microwave frequency was dependent on ionic polarizability as well. As shown in the inset of Fig. 3,

the molecular polarizability ( $\alpha_{-pc}$ ) of  $\text{CoZrTa}_2\text{O}_8$  ceramics was calculated according to the Clausius-Mosotti equation [15]:

$$\alpha_{-pc} = \frac{3V_m}{4\pi} \cdot \frac{K-1}{K+2}$$

where  $V_m$  was the molar volume,  $K$  was the porosity-corrected dielectric constant, which was calculated using the following formula [16]:

$$\frac{\epsilon_r - K}{3\epsilon_r} = \frac{\delta(\epsilon_1 - K)}{\epsilon_1 + 2\epsilon_r}$$

where  $\epsilon_r$  was the measured dielectric constant,  $\epsilon_1$  was dielectric constant of porosity,  $\delta$  was the fractional porosity. It was noted that there was no obvious change in the molecular polarizability under different sintering temperature, implying that sintering temperature had almost no influence to ionic polarizability. The average value ( $\text{avg}(\alpha_{-pc})$ ) of  $\text{CoZrTa}_2\text{O}_8$  ceramics was  $29.37 \text{ \AA}^3$ , which was smaller than the theoretical molecular polarizability calculated as follows:

$$\alpha(\text{CoZrTa}_2\text{O}_8) = \alpha(\text{Co}^{2+}) + \alpha(\text{Zr}^{4+}) + 2\alpha(\text{Ta}^{5+}) + 8\alpha(\text{O}^{2-})$$

where  $\alpha(\text{Co}^{2+}) = 1.65 \text{ \AA}^3$ ,  $\alpha(\text{Zr}^{4+}) = 3.25 \text{ \AA}^3$ ,  $\alpha(\text{Ta}^{5+}) = 4.73 \text{ \AA}^3$ ,  $\alpha(\text{O}^{2-}) = 2.01 \text{ \AA}^3$  [15]. Differences could be found between  $\text{avg}(\alpha_{-pc})$  and  $\alpha_{-theo}$ , suggesting that there was a slight change in ionic polarizability owing to different coordinated environment.

Fig. 4 showed the  $Q \times f$  values and packing fraction of  $\text{CoZrTa}_2\text{O}_8$  ceramics sintered at 1150-1275°C for 4h. The quality factor of ceramics at microwave frequency was mainly caused by both the intrinsic loss as lattice vibration and the extrinsic loss such as the pore, density, grain size and impurities. In this work, the packing fractions were carried out to investigate the intrinsic loss. As shown in Fig.4, the curve of packing fraction presented a downward trend as the sintering temperature increased. It was important to note that the variation trend of  $Q \times f$  values and packing fraction of

CoZrTa<sub>2</sub>O<sub>8</sub> ceramics were inconsistent with each other at the sintering temperature range from 1175°C to 1250°C, implying that some extrinsic factors should be considered to analyze the change of  $Q \times f$  values. Here, grain size, as an extrinsic factor, was considered to be the main factor affected the  $Q \times f$  values. According to the analysis of SEM, when the sintering temperature increased, the grain size also increased while the grain boundary decreased, which resulted in the increasing of  $Q \times f$  values shown in Fig. 4. With further increase of sintering temperature, the  $Q \times f$  values decreased, owing to the abnormal grain growth and packing fraction. In addition, the  $\tau_f$  values of CoZrTa<sub>2</sub>O<sub>8</sub> ceramics changed from -17.1 ppm/°C to -8.7 ppm/°C, and no significant change in  $\tau_f$  values was found under different sintering temperature.

#### 4. Conclusion

High-Q CoZrTa<sub>2</sub>O<sub>8</sub> ceramics were prepared by solid-state reaction and the microwave dielectric properties were investigated. The crystal structure of CoZrTa<sub>2</sub>O<sub>8</sub> ceramics was indexed as wolframite structure belonged to the space group  $P2_1/c$  ( $C_{2h}^4$ ), with only one single-phase being detected from the XRD patterns under different sintering temperature. The dielectric constant of CoZrTa<sub>2</sub>O<sub>8</sub> ceramics was dependent on the relative densities and ion polarizability. The  $Q \times f$  values of CoZrTa<sub>2</sub>O<sub>8</sub> ceramics were affected by grain size and packing fraction. The  $\tau_f$  values of CoZrTa<sub>2</sub>O<sub>8</sub> ceramics had no significant change under different sintering temperature. The promising microwave dielectric properties were reported with  $\epsilon_r=23.54$ ,  $Q \times f=20100$  GHz,  $\tau_f=-8.72$  ppm/°C for CoZrTa<sub>2</sub>O<sub>8</sub> ceramics sintered at 1250°C.

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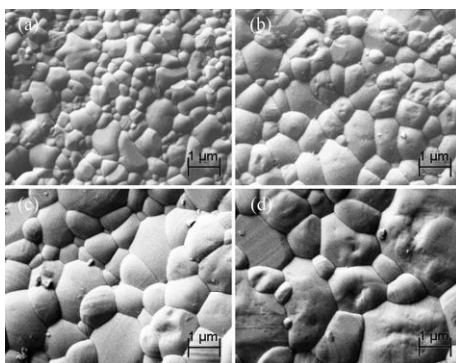


Fig. 1. SEM micrographs of  $\text{CoZrTa}_2\text{O}_8$  ceramics sintered at (a) 1175°C, (b) 1225°C, (c) 1250°C, (d)



1275°C

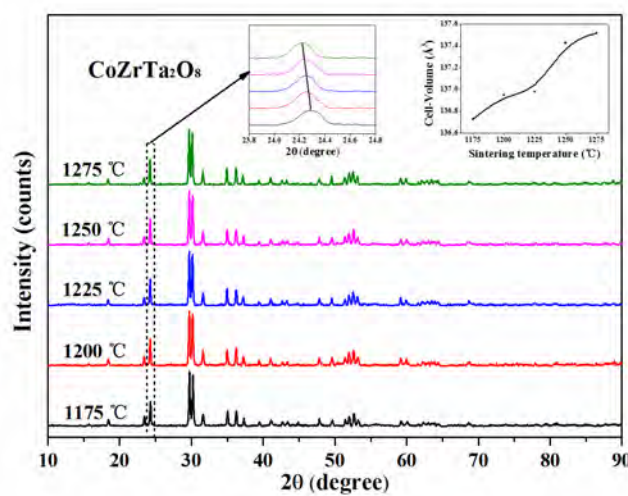


Fig. 2. XRD patterns of  $\text{CoZrTa}_2\text{O}_8$  ceramics sintered at 1175-1275°C for 4h

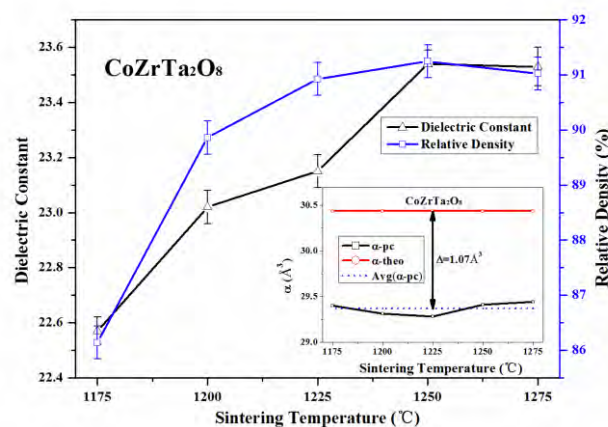


Fig. 3. Dielectric constant, relative density and polarizability of  $\text{CoZrTa}_2\text{O}_8$  ceramics

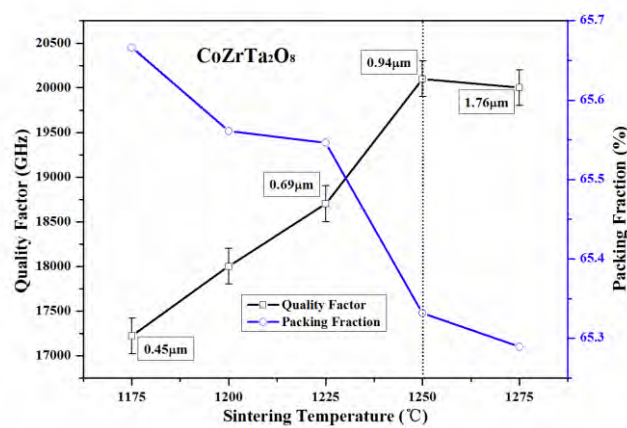


Fig. 4.  $Q \times f$  values and packing fraction of  $\text{CoZrTa}_2\text{O}_8$  ceramics

Highlights:

1. CoZrTa<sub>2</sub>O<sub>8</sub> ceramics are designed for the first time according to ionic substitution on ZnTiNb<sub>2</sub>O<sub>8</sub> ceramics.
2. The new materials of CoZrTa<sub>2</sub>O<sub>8</sub> were successfully prepared by a conventional mixed-oxide route.
3. The crystal structure, ionic polarizability and microwave dielectric properties were investigated.